Pulse and direct current electrodeposition of zinc oxide layers for solar cells with extra thin absorbers

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Abstract: The feasibility of one-dimensional (1D) nanostructured zinc oxide array pulse plating has been presented. An effect of the electrolyte composition, deposition regime and subsequent annealing on structure and optical properties of the electrodeposited ZnO layers has been approved by X-ray diffraction and spectrophotometric analysis. We have determined that for obtaining of ZnO arrays with strong (002) preferable growth orientation in the c-axis direction it is necessary to diminish adsorption of hydrogen and Cl⁻-ions. It has been shown that such conditions are created in electrolyte that contains $0.05 \text{ M Zn}(NO_3)_2$ and 0.1 M NaNO_3 during electrodeposition on FTO-coated glass substrates in pulse plating regime with rectangular impulses of cathode potential (20 ms on-time at $U_{on} = -1.4 \text{ V}$ and 30 ms off-time at $U_{off} = -0.8 \text{ V}$). Therefore, in this work we for the first time have demonstrated the successful growth of 1D ZnO nanostructures by pulse plating without using of templates. The novel electrodeposition technique gives possibilities for the manufacture of the ZnO arrays suitable for solar cells with extra thin absorbers.

Keywords: Electrodeposition, Zinc oxide, Pulse plating

1. Introduction

Zinc oxide (ZnO) has attracted a lot of research interest in recent years due to its unique optical and electronic properties and low cost of materials and fabrication. A wide variety of ZnO crystallite morphologies are observed for both precipitates and thin films including columnar grains, rods, stars and spherical habits [1-4]. Now highly transparent conducting ZnO windows are important components of photovoltaic devices and displays. Recently, solar cells with extra thin absorbers (ETA SC) have shown high potential of ZnO arrays as semiconductor covered electrodes and dye-sensitized photoanodes, particularly, ZnO nanorods proved to be suitable for application in organic photovoltaic devices [5, 6]. An assortment of ZnO nanostructures, such as whiskers, nanowires, nanorods, nanotubes, nanorings and nano-tetrapods have been successfully grown via a variety of methods including chemical vapor deposition, thermal evaporation, and electrodeposition. But despite numerous studies, there is little understanding of the mechanisms and factors that govern the observed morphology [1]. Among other deposition techniques electrodeposition has various advantages, viz. low processing cost, large scale, no vacuum system need, high deposition speed and no use of toxic gases. Effects of electrolyte formula, namely anionic composition [4] and presence of the different organic additives [6], deposition temperature and deposition time [7] and even gravitational level effects [9] on structure and properties of the electrodeposited ZnO nanowire arrays are studied extensively. Nevertheless, there are only rare attempts to employ a pulsed potential technique for ZnO electrodeposition [10]. On the same time, the use of pulse plating is well-known promising way to perfect properties of the electrodeposited layers. That's why purpose of this work is a comparative study of influence of direct current and pulse plating conditions, electrolyte composition and subsequent air annealing on ZnO film structure and optical properties in order to reveal means for obtaining of one-dimensional (1D) zinc oxide nanostructured layers applicable for ETA SC.

2. Methodology

ZnO arrays were electrodeposited on transparent indium tin oxide (ITO) or fluorine doped tin oxide (FTO) covered glass (Pilkington) cathodes in aqueous electrolytes contained ZnSO₄ or Zn(NO₃)₂, KCl and NaNO₃ (Table 1) in three-electrode cell with platinum counter-electrode and saturated Ag/AgCl reference electrode. Electrodeposition of each ZnO layer sample was carried out during 1 hour at 70 °C under potentiostatic conditions (at constant cathode potential U) or under pulse plating regimes with rectangular impulses of cathode potential (20 ms on-time at U_{on} and 30 ms off-time at U_{off}). All potential values in Table 1 are given versus saturated Ag/AgCl reference electrode. In some experiments electrolyte was magnetically stirred (marked + in Table 1). A following treatment of some ZnO layers was fulfilled by air annealing at 200 °C, 300 °C and 400 °C for 1 hour each.

Phase composition and structure of the deposited films were determined by XRD-method using an X-ray diffractometer DRON-4M with CoK_{α} radiation according to θ -2 θ - scheme. Preferable orientations of the films were researched by analytical treatment of the X-ray diffractions by means of obtaining of texture factor P_i [11]:

$$P_{i} = \frac{(I_{i}/I_{0i}) \cdot N}{\sum_{1}^{N} I_{i}/I_{0i}}$$
(1)

where I_i – experimental intensity of maximum; I_{0i} – intensity of this line in accordance with JCPDS card; N – total number of X-ray reflections.

Angles φ between texture axis and surface normal for all reflection planes and P_i values have been calculated according to relation [11]:

$$\cos \varphi = \frac{hh_i + kk_i + \frac{1}{2}(hk_i + h_i k) + \frac{3}{4} \frac{a^2}{c^2} l_i l}{\sqrt{h^2 + k^2 + hk + \frac{3}{4} \frac{a^2}{c^2} l^2} \sqrt{h_i^2 + k_i^2 + h_i k_i + \frac{3}{4} \frac{a^2}{c^2} l_i^2}}$$
(2)

A shape of function $P = f(\phi)$ allow [11] to distinguish degree of texture perfection: the texture is perfect if P decreases rapidly. When the function $P = f(\phi)$ has two or more vertexes, then the structure has two or some texture axes. Average crystalline sizes t (i. e. X-ray domains defined as volumes that diffract coherently) and lattice strains $\Delta d/d$ of the electrodeposited ZnO arrays were determined by the Williamson-Hall formula for adherent deposits [12]. ZnO lattice characteristics a and c were calculated using the formula [11]:

$$\frac{1}{d^2} = \frac{4}{3} \frac{h^2 + hk + k^2}{a^2} + \frac{l^2}{c^2}$$
 (3)

Table 1. Electrolytes	and	electrolysis	regimes	used	for a	denosition	of ZnQ
Tuble 1. Electrolytes	cirici	Ciccii Oiysis	1 CZ III CB	uscu j	OI C	reposition	o, zno.

Sample	Electrolyte	Deposition	(Cathode	e	Magnetic	Current	Charge-
number	r	regime	p	otentia	ıl	stirring	density	area
				(V)			j	ratio
			U	U_{off}	Uon	l	(mA/cm^2)	$q (C/cm^2)$
1	7·10 ⁻⁴ M ZnSO ₄	Potentiosta	tic -1	.3	_	_	+	2→1.4
6.1								
2	0.1 M KCl	Pulse	_	-0.9	-1.5	+	$1.3 \rightarrow 0.5$	3.1
3	0.05 M NaNO ₃	Pulse	_	-1.0	-1.6	+	$2.6 \rightarrow 1.5$	5 5.4
	7·10 ⁻⁴ M ZnSO ₄							
4	0.1 M KCl	Pulse	_	-0.9	-1.5	+	$0.6 \rightarrow 0.4$	1.4
	0.001 M NaNO ₃							
5	$0.05 \text{ M Zn}(NO_3)_2$	Potentiostatic	-1.1	_	_	=	1.1	4.0
6	0.1 M NaNO_3	Pulse		(8.0	-1.4	_	$0.5 \rightarrow 0.8$
2.3								

The transmittance spectra of ZnO layers were measured by double beam spectrophotometer SF-46 in the spectral range $0.4-0.9~\mu m$, when the sample ZnO/FTO/glass was put into working canal and FTO/glass or ITO/glass one was placed in reference canal.

3. Results

As-electrodeposited films were high adherent, semitransparent and scattered visible light. Samples 1 and 2 were grayish in color, but others were white. Figure 1 shows the transmittance spectra (T vs. wavelength λ) for the as-grown and air annealed ZnO layers. As it can be seen, from the one side, the grayish layers increase their transmittance after the annealing (they became white, probably owing to oxidation of Zn traces). From the other side, according to transmittance data, irrespective of electrolyte stirring, sample 3 offers the thinnest near transparent film, samples 1, 2 and 4 were thicker, samples 5 and 6 (not presented in Fig. 1) were the thickest. Assuming ZnO to be typical direct band gap semiconductor, the corresponding optical band gap has been estimated by the zero-crossing of the rising edge of the [(-lnT) x hv]² vs. hv curve [8] (Fig. 1, inset). All obtained band gap values as before such as after annealing correspond to ZnO (E_g near 3.2 - 3.3 eV) [2, 3, 9].

Investigation of structure of zinc oxide arrays electrodeposited in electrolytes and regimes presented in Table 1 has shown (Fig. 2) that all diffraction peaks match the hexagonal structure of wurtzite ZnO (with the exception of reflections assigned to FTO-glass or ITO-glass substrates). Comparative analysis of XRD patterns of the electrodeposited ZnO layers has revealed that, from the one side, the intensity of ZnO diffraction peaks is in direct proportion to concentration of NO₃-ions in the electrolyte that allows us to conclude that thicknesses of ZnO layers grow when amounts of nitrates increase.

From the other side, a deviation of cathode potential towards more negative values (U = -1.3 V for electrodeposited in potentiostatic regime sample 1 and 20 ms on-time at $U_{on} = -1.6$ V and 30 ms off-time at $U_{off} = -1.0$ V for pulse plated sample 3) result in the obtaining of very thin near amorphous ZnO layers, in spite of their large current densities and charge-area ratios of the electrodeposition processes. To our opinion, the reason for that is a most probable intense additive cathode reaction of hydrogen generation in aqueous electrolytes for ZnO deposition that fulfilled at comparatively negative potentials according to relation [13]:

Wavelength (nm)

Wavelength (nm)

$$E^{\circ} = -1.05 \text{ V vs. saturated Ag/AgCl} \qquad (4)$$

$$Sample 1$$

$$Sample 2$$

$$Sample 3$$

$$Sample 3$$

$$Sample 3$$

$$Sample 3$$

$$Sample 40$$

$$Sample 40$$

$$Sample 50$$

$$Sample 50$$

$$Sample 50$$

$$Sample 60$$

$$Sample 60$$

$$Sample 60$$

$$Sample 70$$

$$Sample 70$$

$$Sample 90$$

$$Sample 9$$

Wavelength (nm)

as-grown, —O— annealed at 200 °C, —A— annealed at 300 °C, —— annealed at 400 °C

Fig. 1. O ptical transmittance spectra and the corresponding band gap spectra (insets) of aselectrodeposited and air annealed ZnO arrays.

Probably, hydrogen beads adsorbed on the surfaces of substrates or on the growing ZnO crystals suppress adsorption of Zn^{2+} and OH^- -ions and therefore inhibit growth of zinc oxide arrays, which could be carried out as follows:

$$Zn^{2+} + 2OH^{-} \rightarrow Zn(OH)_{2} \rightarrow ZnO + H_{2}O$$
 (5)

At less negative cathode potentials (U = -1.1 V for electrodeposited in potentiostatic regime sample 5 and U_{on} = -1.5 V and U_{off} = -0.9 V for pulse plated sample 2 and more clearly for sample 6 deposited in the pulse regime at U_{on} = -1.4 V and U_{off} = -0.8 V) current efficiency of the ZnO electrodeposition process increases, that can be seen from comparison of overall intensities of ZnO diffraction peaks for this samples (Fig. 2) and their current densities and charge-area ratios (Table 1).

Required for acceleration of ZnO synthesis cathode reductions of nitrate-ions with creation of OH⁻- groups can be realized according to [13] as follows:

$$NO_3^- + H_2O + 2\bar{e} \rightarrow NO_2^- + 2OH^ E^0 = -0.21 \text{ V vs. saturated Ag/AgCl}$$
 (6)

$$NO_3^- + H_2O + \bar{e} \rightarrow NO_2 \uparrow + 2OH^ E^o = -1.08 \text{ V vs. saturated Ag/AgCl}$$
 (7)

$$NO_3^- + 2H_2O + 3\bar{e} \rightarrow NO\uparrow + 4OH^ E^o = -0.36 \text{ V vs. saturated Ag/AgCl}$$
 (8)

$$NO_3^- + 7H_2O + 8\bar{e} \rightarrow NH_4OH + 9OH^ E^0 = -0.34 \text{ V vs. saturated Ag/AgCl}$$
 (9)

To our opinion, the most useful for ZnO deposition is cathode reaction Eq. (6), because processes Eq. (7) and Eq. (8) produce gaseous compounds whose adsorption can suppress

growth of ZnO, and reaction Eq. (9) is undoubtedly sophisticated multistage process as such as it consumes eight electrons. Therefore, high amount of NO_3^- -ions is a cause of the elevated thicknesses of samples 5 and 6. Enhanced structure of sample 2 as compared with sample 1 is evidently a result of such advantage of pulse electrolysis as suppression of additive cathode reaction Eq. (4), because it is impossible at U_{off} , but during off-time ions NO_3^- and Zn^{2+} can diffuse to the cathode and can be realized processes Eq. (6), Eq. (8) and Eq. (9) which are useful for creation of ZnO.

Moreover, at on-time electrochemical reaction Eq. (7) and reduction of Zn²⁺ are doubtless:

$$Zn^{2+} + 2\bar{e} \rightarrow Zn$$
 $E^{0} = -0.98 \text{ V vs. saturated Ag/AgCl}$ (10)

So, during off-time internal electrolysis is additive possible way for creation of ZnO arrays through following heterogeneous chemical reaction:

$$Zn + NO_3^- + H_2O \rightarrow NO_2^- + Zn(OH)_2$$
 (11)

Table 2 shows structure characteristics of thicker ZnO layers. All ZnO arrays are nanostructured and characterized by little compressive stress (samples 2 and 5) or tension (sample 6). Lattice constants a are near value for single crystal ZnO of hexagonal modification, but the electrodeposited ZnO grains were elongated along c axis (according to JCPDS 36-1451, a = 3.250, b = 5.207).

Table 2. Structure characteristics of the electrodeposited ZnO arrays.

Sample number	Lattice	e constant (Å)	Average crystalline size t (nm)	Lattice strain $\Delta d/d \times 10^4$	
	a	С			
2	3.251	5.228	27	14.5	
5	3.249	5.226	54	3.5	
6	3.253	5.220	16	-36.8	

Comparison of preferable orientations has revealed (Fig. 3) that ZnO layer prepared in electrolyte with low concentration of nitrates (sample 2) has crystallites with random orientation. There seems to be main reason for such structure that the polar (002) crystal plane of the ZnO is capped by Cl⁻-ions (from the KCl supporting electrolyte), which [1, 4] redirect the growth of ZnO. Sample 5 plated at direct current in NO₃⁻-enriched electrolyte has two preferable orientations (002) and (103), probably because of influence of cathode reaction of hydrogen evolution by Eq. (4). Only sample 6 electrodeposited in electrolyte, which contains large concentration of NO₃⁻-ions at pulse plating conditions has strong (002) preferable growth orientation in the c-axis direction. According to [1], increase of (002) reflection in relative intensity is consistent with formation of ZnO rod crystallites along c-axis. In [2-4, 6] judgment, such preferential growth in the (001) plane results in 1D nanostructure of ZnO arrays, e.g. nanowires, nanorods or nanopillars, that grow along the direction perpendicular to the substrate.

4. Discussion and Conclusions

We have determined that for obtaining by electrodeposition of ZnO arrays preferential grown in the (001) plane it is necessary to diminish adsorption of Cl⁻-ions and hydrogen beads on this plane.

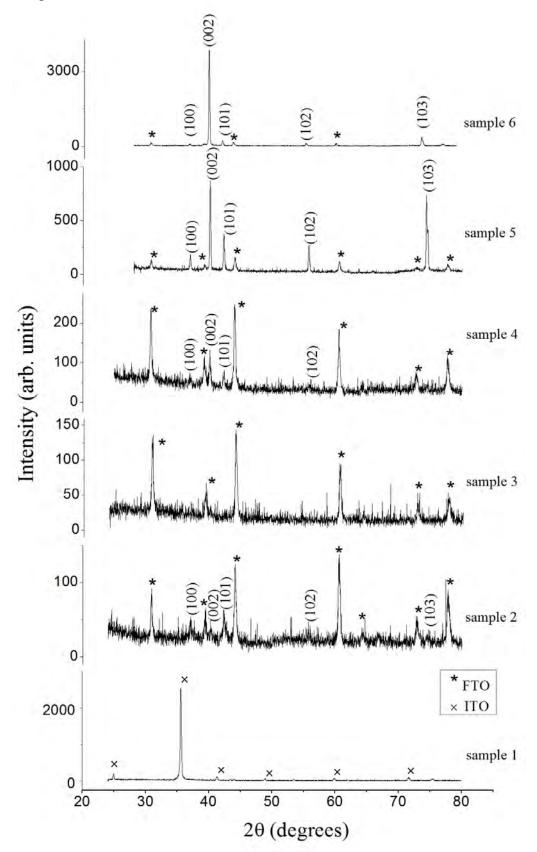


Fig. 2. XRD patterns of ZnO layers electrodeposited onto transparent conducting oxide coating glass substrates (* - FTO, \times - ITO).

It has been shown that such conditions are created in electrolyte that contains 0.05 M $Zn(NO_3)_2$ and 0.1 M $NaNO_3$ during electrodeposition in pulse plating regime with rectangular impulses of cathode potential (20 ms on-time at $U_{on} = -1.4$ V and 30 ms off-time at $U_{off} = -0.8$ V) on FTO-coated glass substrates. Therefore, in this work we for the first time have demonstrated the successful growth of 1D ZnO nanostructures by pulse plating without using of templates. The obtained ZnO arrays have to be the suitable layers for solar cells with extra thin absorbers.

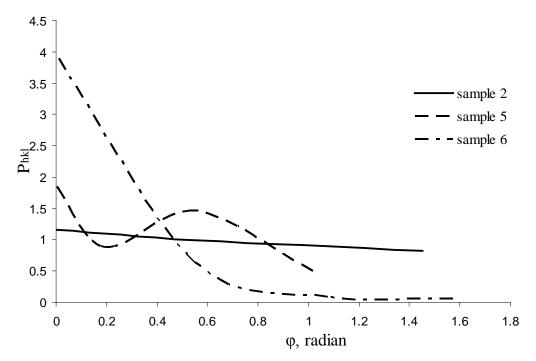


Fig. 3. Degree of texture perfection of electrodeposited ZnO arrays.

Acknowledgments

The work was supported by STCU under Project 4301.

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